

Novel one step hydrothermal synthesis of TiO₂/WO₃ nanocomposites with enhanced photocatalytic activity†

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A novel and simple one step hydrothermal process is used to prepare TiO₂/WO₃ nanocomposites, in which WO₃ and TiO₂ are present in hexagonal and anatase crystalline forms, respectively, and have enhanced photocatalytic activity towards trichloroethylene degradation in the gas phase.

Coupling TiO₂ with other metal oxides (*e.g.*, WO₃ or V₂O₅) is an approach that has received much attention for improving the photocatalytic properties of TiO₂.^{1–8} The absorption of photons with an energy equal to or higher than the TiO₂ semiconductor bandgap (E_g) gives rise to photoexcited electron–hole pairs. Electron–hole pairs can initiate redox reactions with adsorbates at the catalyst surface or can result in their recombination with the release of heat. These two processes compete, and one of the key factors that controls the efficiency of a photocatalyst is the electron/hole recombination rate, as it controls the availability of photoexcited sites on the catalyst surface. When anatase TiO₂ ($E_g = 3.2$ eV) is doped with a semiconductor of smaller bandgap, such as WO₃ ($E_g = 2.8$ eV), improved charge separation can result from the coupling of the two materials.² W(VI) in WO₃ is considered to act as trapping site by accepting the photoexcited electrons from the TiO₂ valence band, hence generating W(V).³ The electron in W(V) can then be transferred to surface reducible species. Moreover, since WO₃ exhibits a higher acidity than TiO₂, its presence in the nanocomposite can modify the affinity of substrates for the catalyst surface, and as a consequence, the adsorption equilibrium and photooxidation activity of the catalyst.³

The addition of WO₃ to TiO₂ has been observed to improve the photooxidation of butyl acetate,⁴ benzene,³ 2-propanol³ and toluene⁵ in the vapor phase. WO₃-doped TiO₂ photocatalysts have, so far, been prepared using only two step procedures by sol–gel,^{6,9} impregnation,^{1,10} microemulsion,⁵ hydrothermal¹¹ and mechanical¹² methods. In all of these synthetic methods, an amorphous precursor is first obtained (step 1), then a thermal treatment at relatively high temperatures (step 2) is carried out to permit the formation of the final photoactive crystalline material.

In this communication, we describe a new synthetic approach based on a hydrothermal process that leads to photoactive TiO₂/WO₃ nanocomposites. This synthesis method is based on a simple “one step” hydrothermal reaction, where TiOSO₄ and ammonium metatungstate ((NH₄)₆(W₁₂O₄₁)·xH₂O) are used as precursors for TiO₂ and WO₃, respectively. In a typical synthesis, appropriate amounts of each precursor were mixed with 100 ml of H₂O and ethanol in a 1 : 1 ratio in a 150 ml autoclave Teflon vessel and reacted hydrothermally at 150 °C for 12 h. The reaction powder obtained was filtered and washed twice with water, ethanol and acetone, then dried at room temperature overnight. The percentage of WO₃ in the nanocomposites was varied between 1 and 40% wt. For comparison purposes, TiO₂ and WO₃ were also prepared by following the same protocol as for the mixed oxides. Each sample was prepared twice to assure consistency in the preparation method.

Fig. 1 shows the XRD patterns of the nanocomposites. The XRD reveals that TiO₂ is present in the anatase form in all samples. The WO₃ has been identified with (NH₄)_{0.33}WO₃ (JCPDS 42-0452) in a hexagonal form, which contrasts with the orthorhombic WO₃ (WO₃·0.33 H₂O) and monoclinic WO₃ obtained by Kominami *et al.*¹³ following hydrothermal treatment at 473 and 523 K, respectively, using a KH₂WO₄ precursor. All other TiO₂/WO₃ nanocomposites exhibit WO₃ in a hexagonal form.

The structural properties of the nanocomposites were measured from duplicate XRD and N₂ adsorption–desorption analyses, and the results are reported in Table 1. The anatase crystallite size was found to be in the range 18–30 nm. The WO₃ loading was found to affect the size of the anatase crystals in the nanocomposites with an

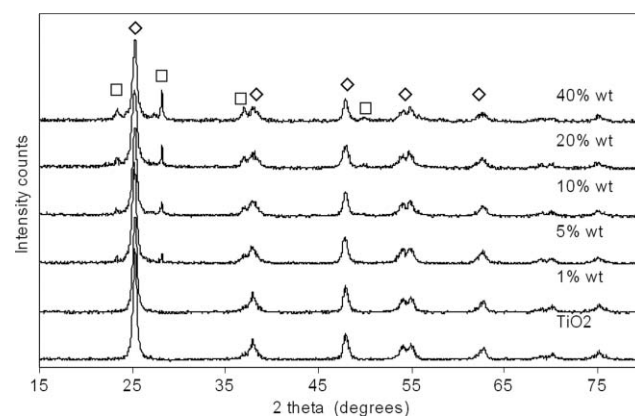


Fig. 1 XRD patterns of TiO₂/WO₃ nanocomposites (□ = WO₃, ◇ = TiO₂).

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† Electronic supplementary information (ESI) available: N₂ sorption isotherms, SEM images and a description of the photocatalytic reactor. See DOI: 10.1039/b711559h

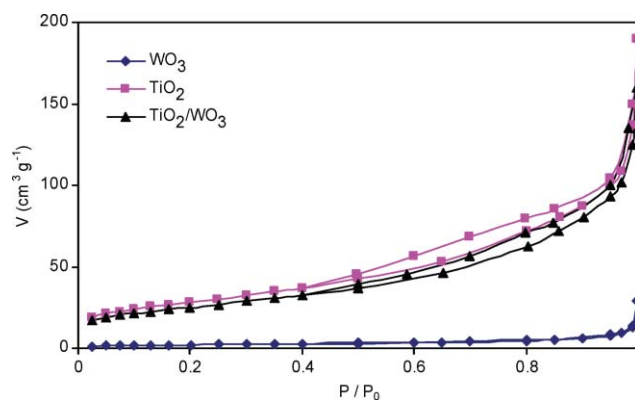
Table 1 Textural properties of TiO₂ samples with different WO₃ contents

WO ₃ (% wt)	<i>D</i> _{anatase} ^a /nm	SSA ^b /m ² g ⁻¹	<i>V</i> _{pore} ^c /cm ³ g ⁻¹
0	30	100	0.30
1	28	86	0.17
5	26	87	0.21
10	23	136	0.22
20	21	102	0.24
40	18	99	0.20
100	—	7.6	—

^a Diameter calculated from the broadening of the (101) anatase peak using Scherrer's equation. ^b Specific surface area determined using the BET method in the range 0.05 < *P/P*₀ < 0.35. ^c BJH desorption pore volume calculated between 2 and 300 nm.

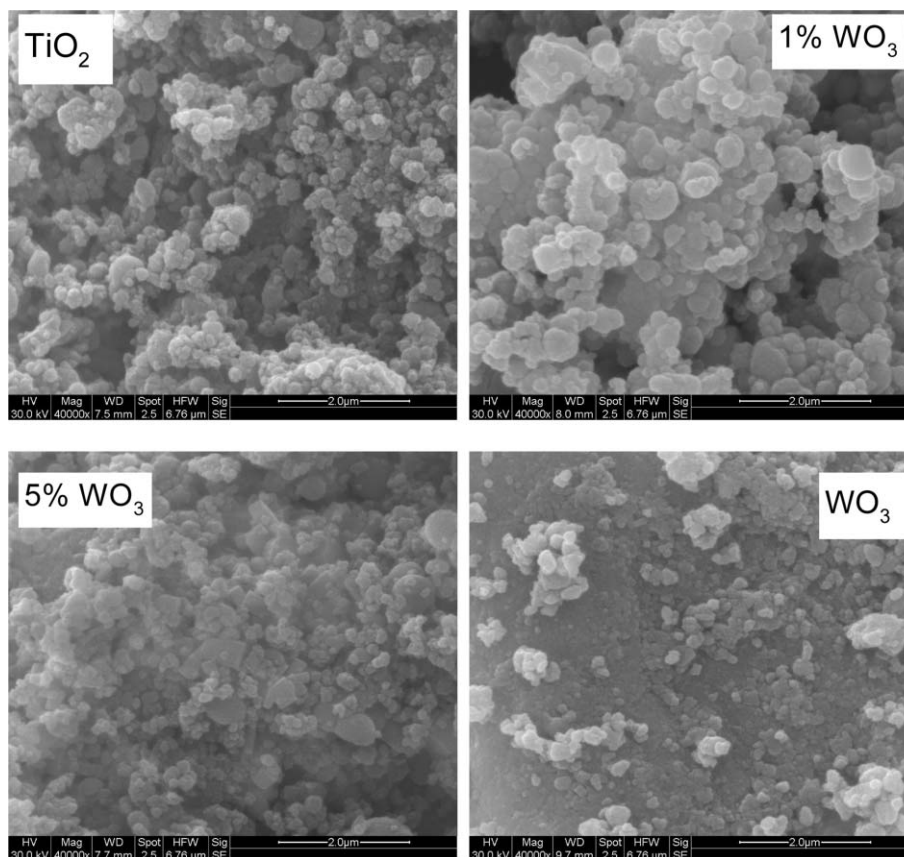
inverse relationship. The WO₃ loading did not significantly affect the porosity properties of the mixed oxides (*i.e.*, pore size and pore volume).

Fig. 2 shows the N₂ sorption isotherm of pure TiO₂, pure WO₃ and the nanocomposite with 20% wt WO₃. Isotherms similar to that of the 20% wt sample were observed for the other mixed oxide nanocomposites (ESI Fig. S1†). The isotherm of pure WO₃ matches that of a typical Type II (BDDT classification), which is usually attributed to non-porous materials or materials of very limited porosity. In contrast, pure TiO₂ and the mixed oxide samples present isotherms indicative of significant porosity. The N₂ sorption isotherms may generally be classified as Type IV, with an inflection and hysteresis loop in the relative pressure *P/P*₀ > 0.3. The isotherms suggest that the porosity of the mixed oxide

**Fig. 2** N₂ adsorption–desorption isotherms of pure Ti and W oxides, and of a TiO₂/WO₃ nanocomposite with 20% wt WO₃.

nanocomposites is very similar to that of pure TiO₂; both types of material have a relatively wide pore size distribution in the mesopore size range. At relative pressures approaching *P/P*₀ = 0.3, the pure TiO₂ and nanocomposites exhibit significant sorption, which we attribute to the presence of large pores or macropores arising from interparticle voids. The higher pore volume of the pure TiO₂ (Table 1) may, in part, be due to the presence of more interparticle voids.

The SEM images shown in Fig. 3 (and ESI Fig. S2†), providing visualisation of the textural properties of large portions of samples, indicate that the pure TiO₂ is made up of irregularly-shaped aggregates with sizes between 0.1 and 1 μm. The particles generally

**Fig. 3** SEM images of pure TiO₂, pure WO₃ and mixed TiO₂/WO₃ samples.

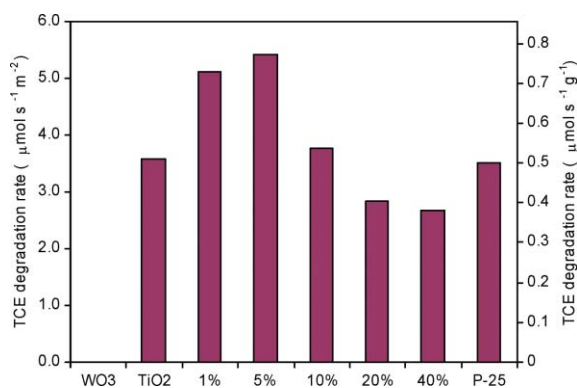


Fig. 4 Reaction rates of TiO₂ samples having different WO₃ contents. Reaction rates are per unit surface area of irradiated coated support (left axis) and per unit weight of supported catalyst (right axis).

exist in loose aggregates with a significant quantity of interparticle voids, which is consistent with the sorption data. Addition of WO₃ does not significantly affect the particle morphology, except for a slight increase in the level of particle aggregation and an associated decrease in interparticle voids. The sample with 10% WO₃ shows a peculiar behavior, with a morphological structure similar to that of pure TiO₂ (ESI Fig. S2†), but with a slight increase in porosity, which corroborates with an increase in surface area (Table 1). Conversely, pure WO₃ exhibits a highly aggregated particle morphology, which may explain the lower level of interparticle voids evident from the sorption data.

The photocatalytic activity of the nanocomposites were evaluated by measuring the rate of decomposition of trichloroethylene (TCE) in air in the presence of water vapour. The nanocomposite powders were deposited on silica glass plates (area 75 cm²) from ethanolic suspensions, achieving a catalyst loading of 0.7 mg cm⁻² for all supports. A narrow slit, flat-plate, single-pass, flow-through photocatalytic reactor was used. A short description and diagram of the reactor is shown in the ESI (Fig. S3).† Further specification details are reported elsewhere.¹⁴ The experiments were conducted in a reaction-controlled regime using a total flow rate of 0.04 L s⁻¹, 8% relative humidity, an initial concentration of TCE in synthetic air (20% O₂, 80% N₂; BOC gases) of 24.8 ± 2 μM and an incident photon flux on the supported catalyst of 5.53 W m⁻² (UVA). Experiments were carried out under “differential reactor” conditions (TCE conversions < 10%) to permit estimation of the reaction rate from the conversion of TCE at steady-state. Reaction rates of the mixed oxide samples were compared to those of single component catalysts evaluated under the same conditions, and with that of commercial TiO₂ Degussa P-25 reference catalyst.

No TCE degradation was detected in the dark or in the absence of catalyst. Therefore, all variation in TCE concentration was ascribed to the photocatalytic activity of the catalyst. The results

are shown in Fig. 4. Pure WO₃ did not show any photocatalytic activity towards TCE under the operating conditions used. The activity of the hydrothermal TiO₂ sample was similar to that of P-25. The nanocomposites with WO₃ contents of 1–5% wt showed up to 42% higher photocatalytic activity compared to that of TiO₂, despite having a lower surface area (Table 1). The enhancement in the rate can therefore be attributed to improved charge separation. However, at WO₃ loadings of 10% wt or higher, the activity reduced as a result of the dilution effect of the inactive WO₃ phase.

In conclusion, the one step hydrothermal method reported here for the preparation of TiO₂/WO₃ nanocomposites in singular crystalline forms resulted in materials with enhanced photoactivity compared to TiO₂ alone. The photocatalytic activities were higher than commercial Degussa P-25 for samples with loadings of up to 5% wt WO₃, possibly due to the coupling effect of the two semiconductors and enhanced charge separation.

To the best of our knowledge, this is the first time that a hexagonal form of WO₃ has been reported in a hydrothermally-prepared nanocomposite that shows enhanced photocatalytic activity. Further work on the optimisation of the synthesis conditions, and a detailed elucidation of the role of the synthesis parameters on the structural properties and photocatalytic activity is currently under way.

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Notes and references

- P. Avila, A. Bahamonde, J. Blanco, B. Sánchez, A. I. Cardona and M. Romero, *Appl. Catal., B*, 1998, **30**, 75.
- G. Marci, L. Palmisano, A. Scalfani, A. M. Venezia, R. Campostrini, G. Carturan, C. Martin, V. Rives and G. Solana, *J. Chem. Soc., Faraday Trans.*, 1996, **92**, 819 and references therein.
- Y. T. Kwon, K. Y. Song, W. I. Lee, G. J. Choi and Y. R. Do, *J. Catal.*, 2000, **191**, 192.
- V. Keller, P. Bernhardt and F. Garin, *J. Catal.*, 2003, **215**, 129.
- A. Fuerte, M. D. Hernández-Alonso, A. J. Maira, A. Martínez-Arias, M. Fernández-García, J. C. Conesa, J. Soria and G. Munuera, *J. Catal.*, 2002, **212**, 1.
- H. Yang, R. Shi, K. Zhang, Y. Hu, A. Tang and X. Li, *J. Alloys Compd.*, 2005, 200.
- A. Rampaul, I. P. Parkin, S. A. O’Neill, J. DeSouza, A. Mills and N. Elliot, *Polyhedron*, 2003, **22**, 35.
- J. Georgieva, S. Armyanov, E. Valova, I. Poullos and S. Sotiropoulos, *Electrochem. Commun.*, 2007, **9**, 365.
- H. Yang, D. Zhang and L. Wang, *Mater. Lett.*, 2002, **57**, 674.
- G. M. Zuo, Z. X. Cheng, H. Chen, G. W. Li and T. Miao, *J. Hazard Mater.*, 2006, **128**, 158.
- D. Ke, H. Liu, T. Peng, X. Liu and K. Dai, *Mater. Lett.*, 2007, DOI: 10.1016/j.matlet.2007.05.060.
- C. Shifu, C. Lei, G. Shen and C. Gengyu, *Powder Technol.*, 2005, **160**, 198.
- H. Kominami, J. I. Kato, S. Y. Murakami, Y. Ishii, M. Kohno, K. I. Yabutani, T. Yamamoto, Y. Kera, M. Inoue, T. Inui and B. Ohtani, *Catal. Today*, 2003, **84**, 181.
- I. Salvado-Estivill, D. M. Hargreaves and G. Li Puma, *Environ. Sci. Technol.*, 2007, **41**, 2028.